A novel chemical reduction route toward fabrication of Fe_3O_4 octahedrons and Fe tubes

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Abstract Novel Fe_3O_4 octahedrons and Fe microtubes were successfully prepared in different alkaline concentration solutions by the reduction of hematite (α -Fe₂O₃) with hydrazine hydrate. The as-prepared powders were characterized in detail by conventional techniques such as X-ray diffraction and field emission scanning electron microscopy. The role of alkaline concentration during the reaction process is discussed in detail. Compared to the reaction in a water system, the reaction in ethanol required less sodium hydroxide and smaller particles were obtained. In addition, the magnetic properties of the samples were characterized using a vibrating sample magnetometer.

Introduction

Magnetic crystals (magnetite, ferrite, iron, cobalt, nickel, etc.) are widely applied in the fields of information storage, color imaging, bioprocessing, magnetic refrigeration, gas sensing, ferrofluids, etc. [1–4]. Numerous methods have been successfully employed to fabricate magnetic crystals and they can be classified into two categories. One is a

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D. Zhang · Z. Tong SORST, Japan Science and Technology Agency (JST), Machida, Tokyo, Japan thermal or sonochemical decomposition of metal-containing complexes [5-7]. The other is chemical reduction which may comprise hydrogen arc plasma [8], the rapid expansion of supercritical fluid solutions (RESS) in the presence of $LiB(C_2H_5)_3H$ [9], a reduction of ferrous ion association colloids with NaBH₄ [10, 11], reduction of hydroxide precursors with hydrogen [12, 13], etc. Among these techniques, chemical reduction is most widely used because of its speed and convenience. However, like the preparation of iron nanocrystals in this method, it remains a challenge because iron salts readily hydrolyze and transform into stable hydroxides which are very difficult to reduce [14]. Boride hydrate is often selected to reduce iron salts in solution but it introduces boride impurities into the product [11]. Among the reductants, hydrazine hydrate has received much attention because of its strong reducing ability, low cost, environmental friendliness, and above all it does not introduce impurities into the system [15]. A variety of metals have been prepared by the reduction of metallic salts with hydrazine hydrate such as copper, silver, ruthenium, nickel, cobalt, iron etc. [16–22].

In this paper, we present a facile method to prepare magnetite octahedrons and iron tubes by reducing hematite $(\alpha$ -Fe₂O₃) with hydrazine hydrate in different alkaline media which helps to avoid the use of toxic materials and complex procedures. Hematite has a rhombohedrally centered hexagonal structure of the corundum type with a close-packed oxygen lattice in which two-thirds of the octahedral sites are occupied by Fe³⁺ ions. Magnetite (Fe₃O₄) crystallizes with a spinel structure. The large oxygen ions are close packed in a cubic arrangement and the smaller iron ions fill in the gaps. The gaps are either a tetrahedral type where iron ion is surrounded by four oxygens. The structural formula for magnetite is [Fe³⁺]A[Fe³⁺, Fe²⁺]BO₄. This particular

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arrangement of cations on the A and B sublattice is called an inverse spinel structure. With negative AB exchange interactions the net magnetic moment of magnetite is due to the B-site's Fe^{2+} . Furthermore, the role of the alkaline concentration during the reaction process is discussed in this paper. The products were found to have different qualities that depend on the ethanol and water solvent mixture.

Experimental section

The chemical reagents used in this work were hematite (Fe₂O₃), hydrazine hydrate (N₂H₄ \cdot H₂O), and sodium hydroxide (NaOH). All the analytical chemicals were purchased from Shanghai Chemical Reagents Company and used without further purification. A typical synthesis proceeded as follows: an aqueous solution was prepared by dissolving 10 mL hydrazine hydrate (N₂H₄ · H₂O), 0.1 g hematite (Fe₂O₃), and different amounts of sodium hydroxide (NaOH) in 30 mL distilled water. After vigorous stirring with a magnetic stirrer bar at room temperature for 10 min, the solution was transferred to a Teflon-lined stainless steel autoclave with a 50 mL capacity. The autoclave was sealed and the solution was kept at 160 °C for 24 h. Upon completion of the reaction, the autoclave was allowed to cool to room temperature. The resultant black solid precipitates were then easily separated from the reaction system by a permanent magnet. The supernatant was discarded by decantation. Distilled water and alcohol were used to wash the precipitates, and this was done several times. Finally, the black products were dried at 80 °C in a vacuum oven.

Characterization

The as-synthesized samples were characterized by X-ray diffraction (XRD, Philips X'Pert PRO SUPER X-ray diffractometer with Cu K α radiation ($\lambda = 0.154178$ nm), field emission scanning electron microscopy (FE-SEM, JEOL JSM-6700M), and with a vibrating sample magnetometer (VSM, BHV-55).

Results and discussion

It is well known that hydrazine hydrate has a strong reducing ability:

$$N_2 + 4H_2O + 4e^- = N_2H_4 + 4OH^-, \Phi^0 = -1.15V$$

Therefore, the amount of alkali is a key factor that influences the reaction process. The products obtained by using different amounts of alkali during the reaction process were



Fig. 1 XRD patterns of the samples produced using different amounts of alkaline: a, b, c, d: in the water system; e: in the ethanol system (a) 0; (b) 1 g; (c) 10 g; (d) 20 g; (e) 5 g

analyzed by XRD. Without using any alkali the only product was the starting reagent Fe₂O₃ (Fig. 1a) which indicates that no reaction took place. Figure 1b shows that the product obtained in the solution containing 1 g of alkaline was Fe₃O₄. With 10 g of alkali, the characteristic peaks of α -Fe appeared and the product was a mixture of α -Fe and Fe₃O₄ (Fig. 1c). Using 20 g of alkali, a pure α -Fe phase formed (Fig. 1d). At a high alkali concentration the reducing ability of hydrazine was greatly enhanced. Consequently, the reaction between hydrazine and Fe(III) proceeded far more easily. The fact that iron was obtained at a comparatively high alkali concentration further suggests that the alkali concentration is important for the formation of iron.

When water was replaced by ethanol as solvent and with a constant amount of hematite the formation of iron was easier and required less alkali. Figure 1e gives the XRD pattern of the sample produced with 5.0 g sodium hydroxide and this pattern is assigned to a pure α -Fe phase. This phenomenon was also observed in the work of Zheng et al. [23, 24] where they prepared nickel and cobalt nanocrystals using a similar method. This behavior was attributed to the effect of the solvent as it caused the reaction to be more moderate. In water, the precursor of Fe(OH)₃ stabilizes easily and is then difficult to reduce [14]. In the case of ethanol, the activity of Fe(OH)₃ was almost constant in the presence of ethanol which resulted in milder reaction conditions.

Sizes and morphologies of the Fe_3O_4 samples were studied using FESEM. Figure 2 shows representative FESEM images of the sample prepared in solution with 1 g of alkaline. The low-magnification image in Fig. 2a shows that the magnetite product is composed of octahedrons with





sizes ranging from 2 to 5 µm. These particles display a narrow size distribution as shown in the FESEM image. A high-magnification FESEM image of the magnetite microoctahedrons is shown in Fig. 2b. These magnetite octahedrons have sharp edges and smooth facets which further indicate that the particles are well crystalline and this is in agreement with the XRD pattern. The morphology of a crystal is mostly determined by the relative growth rate between different planes during its formation. Wang suggested that the morphology of an fcc crystal is determined by the ratio of the growth rate in the $\langle 100 \rangle$ direction to that in the $\langle 111 \rangle$ direction [25]. Octahedrons bounded by eight {111} planes will be formed when the growth rate along the $\langle 100 \rangle$ direction is faster than that along the $\langle 111 \rangle$ direction. In our system, the concentration of Fe^{3+} ions was very low in solution because the solubility of Fe₂O₃ particles is very low. Figure 3 shows FESEM images of Fe₂O₃ at different stages during the reaction. Figure 3a shows that the Fe₂O₃ product mainly consists of plates of about 1 μ m in size at the start of the reaction. At a reaction time of 3 h, the Fe₂O₃ mainly consisted of nanoparticles that were 200-300 nm in size as shown in Fig. 3b. This figure indicates that Fe₂O₃ was slowly dissolved during the reaction. Therefore, a lower concentration of raw materials leads to a slower reaction rate. In addition, the reaction rate of Fe^{3+} ion reduction to Fe²⁺ ion was slow which contributed to the low formation rate of Fe₃O₄. The slow formation rate is advantageous for a faster growth rate along $\langle 100 \rangle$ compared to that along $\langle 111 \rangle$. Theoretically, however, the magnetite crystal grows as an octahedron because $\{111\}$ surfaces have the lowest energy. Slow reaction and growth rates cause magnetite to grow in its typical crystalline habit [26]. At reaction times of 72 h or more the morphology of the Fe₃O₄ samples remained octahedron-like. This indicates that the end shape represents the equilibrium crystal shape.

Figure 4 shows FESEM images of the two Fe samples that were obtained in water and ethanol, respectively. Particles obtained from both the water and ethanol systems were of the same hollow rod shape. However, the particles from the water system (Fig. 4a, b) had sizes ranging from 100 to 120 µm while the particles from the ethanol system (Fig. 4c, d) had sizes ranging from about 10 to 50 µm. It is well known that small metal particles dissolve easily in hot alkali solutions. Therefore, we deduce that an eroding effect of alkali on iron crystals is present. In the water system, the small iron particles dissolved easily because of the higher alkali concentration. In the presence of excess reductant, dissolved particles in the water system could be reduced again. Because of Ostwald ripening, newly produced iron atoms transfer onto the surfaces of larger crystals that have high surface free energies [27–29]. As a result, the final product was relatively larger in size. In ES, however, the eroding effect was much weaker because of the decreased alkali concentration and this contributed to the smaller particle size.

Magnetic measurements on the as-prepared octahedral magnetite were conducted and the M-H hysteresis loop is

Fig. 3 SEM images of the reaction system with 1 g of alkaline at different times: **a** 0 h, **b** 3 h



Fig. 4 a low-magnification and b high-magnification FESEM image of the iron sample prepared in the water system with 20 g of alkaline; c low-magnification and d high-magnification FESEM image of the iron sample prepared in the ethanol system with 5 g of alkaline



Fig. 5 M-H (magnetizationhysteresis) loops of as-prepared samples measured at room temperature: **a** Fe_3O_4 octahedrons, **b** Fe tubes prepared in the water system

shown in Fig. 5a. The hysteresis loop of the products shows ferromagnetic behavior with saturation magnetization (M_s) , remanent magnetization (M_r) , and coercivity (H_c) values of about 81, 22 emu/g and 190 Oe, respectively. The saturation magnetization (M_s) value is different from those reported for Fe₃O₄ nanowires (68.7 emu/g) [30] and nanoparticles (85.8 emu/g) [31]. This may be due to the low shape anisotropy and the multiple domains of Fe₃O₄ octahedrons which facilitate magnetization in the directions along their easy magnetic axes. Many factors such as size, structure, surface disorder, morphologies, etc. may influence the magnetic properties of materials. Anisotropy, including crystal anisotropy and shape anisotropy, caused the lower saturation magnetization observed in this work. This plays an important role in reducing the saturation

anisotropy. The H_c and the M_s values are opposite to those expected from the Stoner–Wohlfarth theory: $H_c = 2 K/(\mu_0 M_s)$, where K is the magnetocrystalline anisotropy constant and μ_0 is the vacuum permeability [32]. Therefore, the higher H_c leads to a lower M_s for the samples with cubic magnetocrystalline anisotropy. Enhanced anisotropy induces a large magnetic coercivity where the magnetic spins are preferentially aligned along the long axis and their reversal to the opposite direction requires higher energies than for lower shape anisotropy [33]. The order of shape anisotropy was found to be nanowires > octahedrons > nanoparticles. As the shape of anisotropy increases, the H_c increases. When the H_c increases the M_s decreases based on the above considerations.

magnetization because most materials contain some type of

We also measured the magnetization of the iron samples that were obtained in water. Figure 5b gives their magnetization hysteresis (M-H) loops. The coercivity was 197 Oe which is lower than the reported values for Fe nanorods (900 Oe) [34]. The lower coercivity obtained in this work may be caused by lower anisotropy, which plays a vital role in determining the coercivity of magnetic crystals.

Conclusions

In summary, we prepared novel Fe_3O_4 octahedrons and Fe tubes in different concentrations of alkaline solution by reducing hematite with hydrazine hydrate. Using a high amount of alkaline (20 g), pure iron was produced. By comparison with the reaction in water, the reaction in ethanol required less sodium hydroxide and smaller particles were obtained.

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